

Amorphous-Phase-Mediated Crystallization of Ni Nanocrystals Revealed by High-Resolution Liquid-Phase Electron Microscopy

Jiwoong Yang,^{†,‡,§,∥} Jahyun Koo,[⊥] Seulwoo Kim,[‡] Sungho Jeon,[#] Back Kyu Choi,^{†,‡} Sangwoo Kwon,[‡] Joodeok Kim,^{†,‡} Byung Hyo Kim,^{†,‡} Won Chul Lee,[#] Won Bo Lee,[‡] Hoonkyung Lee,[⊥] Taeghwan Hyeon,*,^{†,‡} Peter Ercius,*,[∥] and Jungwon Park*,^{†,‡}

Supporting Information

ABSTRACT: Nonclassical features of crystallization in solution have been recently identified both experimentally and theoretically. In particular, an amorphous-phasemediated pathway is found in various crystallization systems as an important route, different from the classical nucleation and growth model. Here, we utilize highresolution in situ transmission electron microscopy with graphene liquid cells to study amorphous-phase-mediated formation of Ni nanocrystals. An amorphous phase is precipitated in the initial stage of the reaction. Within the amorphous particles, crystalline domains nucleate and eventually form nanocrystals. In addition, unique crystallization behaviors, such as formation of multiple domains and dislocation relaxation, are observed in amorphousphase-mediated crystallization. Theoretical calculations confirm that surface interactions can induce amorphous precipitation of metal precursors, which is analogous to the surface-induced amorphous-to-crystalline transformation occurring in biomineralization. Our results imply that an unexplored nonclassical growth mechanism is important for the formation of nanocrystals.

rystallization in liquid media is a ubiquitous phenomenon ✓ and is fundamental to understanding the formation of many materials, including colloids, biominerals, and nanocrystals. In the classical theory of colloidal chemistry, crystallization is described by nucleation and growth, where it is assumed that lattices are formed by ion-by-ion addition.^{1,2} Recently, nonclassical features of crystallization have been identified.3-5 In particular, amorphous phases formed at the initial stage of crystallization are known to provide an important free-energy landscape for materials formation.^{3,5c-e,6} Amorphous-phase-mediated crystallization processes have been mainly observed in biomineralization processes, such as the formation of proteins^{3d,e} and calcium phosphate/ calcite. 5c,d,6 However, many questions regarding its interplay with classical pathways and generality in different materials are

less explored and await experimental approaches. A major difficulty in the study of this phenomenon originates from the lack of characterization methods for direct observation of the process with high spatial and temporal resolution. The recent development of liquid-phase in situ transmission electron microscopy (TEM) provides new capabilities for the direct and real-time observation of material dynamics in liquid media, 5,7,8 which has not been accessible by previous characterization tools. Combining this method with the technical advances in TEM, such as aberration-corrector optics, ¹⁰ enables the *in situ* image acquisition with high resolution (HR). Thus, it becomes possible to resolve different phases and crystallinity of materials in liquid.

Here, we present the direct HR observation of amorphousphase-mediated Ni nanocrystal (NC) formation by graphene liquid cell (GLC) TEM. We encapsulated a molecular precursor solution for Ni NC formation in GLCs for in situ TEM observation. Atomic-resolution TEM imaging shows distinct stages of Ni NC formation. An amorphous phase is rapidly aggregated from the homogeneous solution. The nucleation and growth of crystalline domains then drive the formation of Ni NCs. The experimental result is supported by theoretical calculations. Interestingly, this process is similar to the surface-mediated amorphous-to-crystalline transition of biominerals in a reduction-reaction-limited manner. In addition, multiple nucleations of crystalline phase in a single amorphous aggregate and relaxation of dislocations at grain boundaries are also observed, which highlights the diversity in the NC formation pathways.

We imaged Ni NC formation in GLCs¹¹ using TEM with chromatic and spherical aberration correction (see Figure S1 and sections 1.1-1.3 in the Supporting Information for experimental details). A homogeneous Ni(II) growth solution containing Ni-ammine-acetate complexes (Figure S2) was prepared and encapsulated in GLCs. Using GLCs minimizes

Received: November 7, 2018 Published: January 4, 2019



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[†]Center for Nanoparticle Research, Institute for Basic Science (IBS), Seoul 08826, Republic of Korea

^{*}School of Chemical and Biological Engineering, Institute of Chemical Processes, Seoul National University, Seoul 08826, Republic of Korea

 $^{^{8}}$ Materials Sciences Division and $^{\parallel}$ Molecular Foundry, Lawrence Berkeley National Laboratory, Berkeley, California 94720, United

¹Department of Physics, Konkuk University, Seoul 05029, Republic of Korea

Department of Mechanical Engineering, Hanyang University, Ansan, Gyeonggido 15588, Republic of Korea

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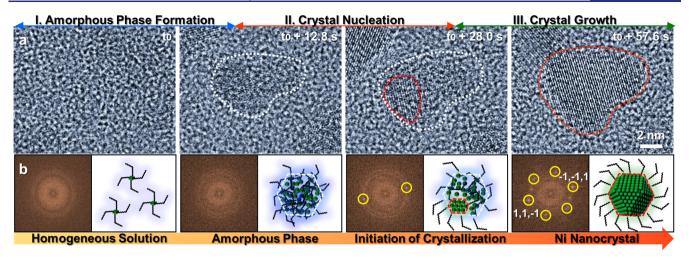


Figure 1. Overall view of amorphous-phase-mediated formation of Ni NCs. (a) Representative TEM images showing the reaction stages of amorphous-phase-mediated crystallization: (i) amorphous-phase formation, (ii) crystal nucleation, and (iii) crystal growth. The white and red dashed lines represent the amorphous and crystalline phases, respectively. (b) The corresponding FFT images and schematic illustrations.

undesired scattering of the electron beam. 11,12 As a result, the direct observation in GLCs secures a sufficiently high contrast and spatial resolution for observing the first-row transition metal in solution, otherwise obscured by background noise. Low-resolution in situ TEM (Figure S3 and Movie S1) presents an overview of the Ni NC formation. The formation of Ni NCs is confirmed by TEM images, electron energy loss spectroscopy, and energy dispersive X-ray spectroscopy (Figure S4). Nanoparticles (NPs) are recognized by the evolution of dark dots, and they grow to sub-10-nm.

HR in situ TEM observation shows a temporal view of individual NC formation (Figure 1 and Movie S2). At the early stage, an amorphous phase precipitates from the homogeneous solution. Within the domain marked with white dashed lines (Figure 1a), aggregated intermediates show randomly packed granular features (Movies S2 and S3). TEM simulations support that the domain does not contain crystalline species (Figures S5 and S6).¹³ A crystalline domain (mapped by red dashed lines) nucleates from the corner and can be clearly identified by the lattice fringe formation (Figure 1a). Crystalline domains then grow gradually, whereas the remainder of the NP maintains an amorphous phase with the slight increase of overall NP size. Eventually, the entire domain transforms into the face-centered cubic (fcc) crystalline phase of Ni, 14 which is clearly identified from bright-field images and fast Fourier transform (FFT) patterns (Figure 1b). This type of growth pattern is different from the classical growth mechanism of colloidal NCs under diffusionlimited growth conditions. Conventionally, NCs nucleate stochastically at the initial stage, followed by an increase of their size throughout the growth due to gradual monomer attachment and/or coalescence events.^{2,15} Additional HR in situ (Figure S7 and Movie S3) and wide-view movies (Movie S4) present similar formation pathways, suggesting that the amorphous-phase-mediated crystallization of Ni NCs occurs ubiquitously. Seeing the burst of NC formation in a lowmagnification movie, it is likely that the formation of amorphous phases occurs relatively quickly at the earlier stage, and the onset of amorphous-to-crystalline transformation takes place stochastically over a prolonged time period.

To gain a better understanding of the process, we analyze the growth of the crystalline domains in more detail. To locate crystalline domains at different time frames (Figure 2a,b), we perform masking and inverse FFT of bright-field TEM images (Figure 2c,d). This enables us to trace the growth kinetics of the crystalline domains (Figure 2e,f). The analysis of the other NC from Movie S3 shows consistent results (Figures S8 and S9). It has been known that one of the major reaction pathways for the amorphous-to-crystalline transformation in biomineralization is induced by surface interactions. 6 It is likely that the amorphous phase observed in our experiments is an amorphous intermediate precipitated on the graphene surface. This is supported by the fact that the translational and rotational motion of NCs is significantly suppressed during their growth (Figure 2c,d). It is also well known that bulk Ni metal strongly interacts with graphene. 16 These facts suggest a hypothesis that the process we observed is comparable to the surface-induced amorphous-to-crystalline transformation. Interestingly, the detailed reaction mechanism can be adopted from the Ni electroplating process, 1

$$[\operatorname{Ni}(L)_{6}]_{\text{free}}^{2+} + L_{\text{surf}} \stackrel{k_{1}}{\underset{k_{2}}{\rightleftharpoons}} [\operatorname{Ni}(L)_{6}]_{\text{surf}}^{2+} + L_{\text{free}}$$
(1)

$$[\text{Ni}(\text{L})_6]_{\text{surf}}^{2+} + 2\text{e}^{-\frac{k_3}{2}} \text{Ni}(\text{s}) + \text{L}_{5,\text{free}} + \text{L}_{\text{surf}}$$
 (2)

where L is an organic ligand molecule. The precipitation of aggregates on the graphene surface and the subsequent crystallization are analogous to reactions (1) and (2), respectively. The reduction reaction is presumably the rate-determining step of the crystal generation, because the amorphous intermediates are rapidly formed at the initial stage. Thus, $\mathrm{d}V/\mathrm{d}t$ is approximately proportional to $k_3\{[\mathrm{Ni}(\mathrm{L})_6]_{\mathrm{surf}}^{2+}\}^{\mathrm{x}}[\mathrm{e}^-]^{\mathrm{y}}$, which is almost constant since the concentration of Ni complexes in aggregates and the electron dose rate are unchanged. The crystal growth rate $(\mathrm{d}V/\mathrm{d}t)$ is measured as a constant for each NC, and the growth is completed earlier at the higher dose rate (Figure 2e,f and Figure S9), which is consistent with our suggestion.

Furthermore, the suggested formation mechanism is supported by theoretical calculations. Molecular dynamics (MD) calculations show that Ni²⁺ ions tend to form aggregates

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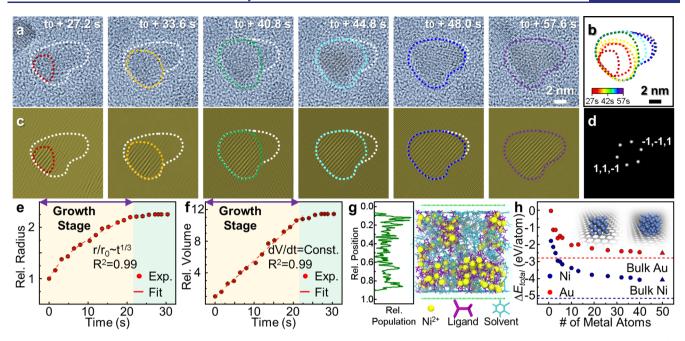


Figure 2. Crystalline-phase formation. (a) A time series of TEM images showing the growth of the crystalline phase from the amorphous phase, (b) the corresponding contour plot, and (c) inverse FFT images. The corresponding movie is displayed in Movie S2. (d) FFT of the last image frame. (e,f) Growth kinetics of the crystalline domain shown in time-dependent change of the relative radius (e) and volume (f). The first frame at t_0 + 27.2 s is set as the standard for plotting. (g) MD calculations showing the distribution of Ni²⁺ ions in GLC. (h) DFT calculation estimating the formation energy of Ni (blue) and Au (red) NPs as a function of the number of metal atoms, where dots and triangles denote the crystalline and amorphous NPs, respectively. The blue and red dashed lines represent the formation energy of bulk Ni and Au crystals. The inset images show the structure models for Ni₁₉ and Ni₅₀.

on the graphene surface (Figure 2g and Figure S10). We also estimate the formation energy of NPs on graphene as a function of the number of atoms by density functional theory (DFT) (Figure 2h). The atomic structures of the NPs used for calculations are displayed in Figure S11. The formation energy of Ni NPs decreases as the size of the NPs increases. Surprisingly, the amorphous Ni NPs on graphene (50 atoms in Figure 2h) show formation energy similar to that of crystalline ones, indicating that substrate interaction contributes to stabilizing amorphous phases. In addition, the surface energy of an amorphous NP is lower than that of a crystalline NP, which helps to expedite the formation of amorphous intermediates (Table S1). It is noteworthy that the formation energy of Ni NPs on graphene is much lower than that of Au NPs. Previous reports show that noble metal NCs typically grow by the classical nucleation-and-growth model. This implies that interactions with a substrate can guide an alternative growth pathway, such as the rapid amorphous condensation and subsequent crystallization process seen in the reported experiments.

As nucleation of the crystalline domain is stochastic, such events may occur at multiple sites within a single amorphous aggregate. Our HR *in situ* TEM imaging shows the sequential nucleation of multiple crystalline grains in the one amorphous aggregate (Figure 3a and Movie S5). The corresponding Fourier-filtered images mapping these domains are displayed in Figure S12. In the initial stage, the first crystalline domain, marked with the red dashed line, is formed at the local region of the amorphous aggregate and expands. Another crystalline domain then appears in a different region, which is indicated by the yellow dashed line (Figure 3a). The nucleation and propagation of the last crystalline domain, marked with the blue line, follow in the later stage (Figure 3a). Consequently,

the resulting Ni NC in the last image, at t_0 + 201.6 s, shows the coexistence of three crystalline domains with sharp interfaces. We readily find populations of both single-crystal and polycrystalline NCs from the same experiment (Figure S13), highlighting diversity in the crystallization pathways. Coalescence of small NCs along the specific crystal direction that minimizes the entropic barrier has been suggested as one of the major growth mechanisms of colloidal metal NCs with multiple domains. Sa,11a,b Our direct observation implies that the formation of multiple crystalline grains in the amorphous-phase-mediated crystallization can be an alternative pathway for the formation of multigrained NCs.

Furthermore, we observe the relaxation of a dislocation in a polycrystalline NC to form a single-crystalline NC (Figure 3be and Movie S6). The dislocation in Figure 3b,c is formed by two crystalline domains with a large difference in their sizes. The boundary is intrinsically curved and kinked. Such a dislocation is presumably thermodynamically unstable. During the relaxation, the kink in the first image of Figure 3b,c, at t_0 , is first removed by the fast incursion of the dominant domain into the region near the kink; thus, the boundary develops a smooth curvature. The grain boundary is, then, gradually expelled until it completely leaves the NC (Figure 3e). The two domains keep a sharp interface and the relative crystal orientation during the movement of the grain boundary (Figure 3d). The NC also maintains its overall size and shape. Considering that grain boundaries in other polycrystalline NCs (Figure 3a, Movie S6, and Figure S13) are well preserved, even at the higher dose rate, the relaxation process may be mainly induced by the small misorientation and the large size disparity between adjunct crystal domains. 11b

In summary, we report the direct observation of amorphousphase-mediated crystallization of Ni NCs using GLC TEM.

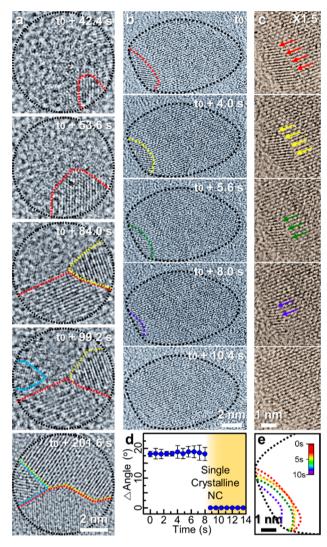


Figure 3. Multiple nucleation and dislocation relaxation. (a) A time series of TEM images showing multiple nucleation. The colored lines highlight three crystalline domains. The corresponding movie is displayed in Movie S5. (b) A time series of TEM images of dislocation relaxation. The dashed lines highlight the grain boundaries. The corresponding movie is displayed in Movie S6. (c) Magnified TEM images around the boundary without the dashed lines. (d) Relative misorientations between two crystal domains in panel (b). (e) Contour plot showing the dislocation relaxation.

Our results uncover an unexplored reaction pathway of NC synthesis and highlight the diversity in crystallization processes.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/jacs.8b11972.

Movie S1: low-resolution movie showing the Ni NC formation (AVI)

Movie S2: high-resolution movie showing the amorphous-phase-mediated NC formation (AVI)

Movie S3: high-resolution movie showing the amorphous-phase-mediated NC formation (AVI)

Movie S4: wide-view movie showing the amorphousphase-mediated NC formation (AVI)

Movie S5: high-resolution movie showing multiple nucleation in a single amorphous aggregate (AVI)

Movie S6: high-resolution movie showing dislocation relaxation (AVI)

Methods and additional TEM and spectroscopy data, including Figures S1–S13 (PDF)

AUTHOR INFORMATION

Corresponding Authors

- *thyeon@snu.ac.kr
- *percius@lbl.gov
- *jungwonpark@snu.ac.kr

ORCID

Byung Hyo Kim: 0000-0002-4098-0053 Won Chul Lee: 0000-0001-8479-0836 Won Bo Lee: 0000-0001-7801-083X Hoonkyung Lee: 0000-0002-6417-1648 Taeghwan Hyeon: 0000-0001-5959-6257 Peter Ercius: 0000-0002-6762-9976 Jungwon Park: 0000-0003-2927-4331

Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

This work was supported by IBS-R006-D1 (T.H. and J.P.); the National Research Foundation (NRF) of Korea, funded by the Korea government (MSIT) (No. NRF-2017R1C1B2010434 and No. NRF-2017R1A5A1015365) (J.P.); the MOTIE (Ministry of Trade, Industry & Energy) and KRSC (Korea Semiconductor Research Consortium) support program for the development of future semiconductor devices (No. 10080657) (J.P.); and the U.S. Department of Energy (DOE), Office of Science, Office of Basic Energy Sciences, Materials Sciences and Engineering Division, under Contract No. DE-AC02-05CH11231 within the KC22ZH program (J.Y. and P.E.). Work at the Molecular Foundry was supported by the Office of Science, Office of Basic Energy Sciences, of the U.S. DOE under Contract No. DE-AC02-05CH11231. The Institute of Engineering Research at Seoul National University provided research facilities for this work. This work was also supported by the Basic Science Research Program (No. NRF-2018R1D1A1B07046751) through the NRF of Korea, funded by the Ministry of Education, Science and Technology (J.K. and H.L.); the Supercomputing Center/Korea Institute of Science and Technology Information with supercomputing resources including technical support (KSC-2018-C2-0024) (S.K. and W.B.L.); the Basic Science Research Programs through the NRF of Korea funded by the Ministry of Science and ICT (2016R1C1B1014940); and the Ministry of Education (2018R1D1A1B07050575) (J.S. and W.C.L.).

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